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Enhanced photocatalytic activity of palladium decorated TiO₂ nanofibers containing anatase-rutile mixed phase

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ARTICLE INFO

Article history:

Received 24 October 2014

Received in revised form

29 December 2014

Accepted 2 January 2015

Available online 4 March 2015

Keywords:

TiO₂

Nanostructures

Photocatalyst

Photodegradation

Photocatalytic hydrogen generation

ABSTRACT

Titanium dioxide (TiO₂) based material is probably the most promising environmentally friendly photocatalyst, with low cost, high photocatalytic activity, and excellent photostability as demonstrated in photocatalytic hydrogen generation and in abatement of organic pollutant. In this study, sodium hydrogen titanate nanofibers were prepared by alkaline hydrothermal synthesis at first. Then, a quick screening was carried out to find the appropriate calcination condition for preparing the high catalytically active titanium dioxide nanofibers (TiO₂ NFs) with anatase-rutile mixed phase (anatase-rutile TiO₂ NFs). The crystalline structure of anatase-rutile TiO₂ NFs was analyzed by Raman spectrometer and X-ray diffraction meter. The anatase-rutile TiO₂ NFs calcined at 800 °C for 4 h shows the highest decomposition rate of the brilliant green, higher than that of the commercial photocatalyst – Degussa P25. Furthermore, palladium based nanoparticles was decorated on the surface of anatase-rutile TiO₂ NFs to enhance the photocatalytic hydrogen performance. We also developed a method to make use of catalyst materials for large-area coatings and freestanding films by preparing the cellulose/catalyst composite films. The obtained catalysts in this study might be a reasonable alternative for the commercial TiO₂, Degussa P25, for photocatalytic applications in the decomposition of organic dyes and photocatalytic hydrogen generation.

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Introduction

In recent years, energy and environmental issues are significant topics around the world. The discovery of photoelectrochemical splitting of water on titanium dioxide (TiO₂) by Fujishima and Honda in 1972 has initiated a considerable boom of semiconductor-based photocatalyst research [1]. TiO₂

is the subject of intensive research as the most promising photocatalyst because of its good photocatalytic activity and excellent photostability, at the same time being environmentally friendly and low cost [2,3]. In addition, TiO₂ can also be used to address several aspects of modern renewable energy production, such as efficient solar cell electrode [4,5], photocatalytic hydrogen production [6–8], and management of environmental pollution [9,10]. TiO₂ is a wide-band-gap

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<http://dx.doi.org/10.1016/j.ijhydene.2015.01.012>

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semiconducting material with several natural crystalline phases, such as anatase, rutile, brookite, and etc. [11,12] Actually, some literatures demonstrated that the photocatalytic activity of anatase/rutile mixed phase is higher than single phase (anatase or rutile) due to the formation of barrier between the anatase and rutile surface to enhance the electron–hole separation rate [13–16]. The alkaline hydrothermal synthesis has opened new possibilities for large scale and simple production of various types of titanate nanostructures such as nanoparticle, nanofiber, and nanotube [17,18]. The titanates can be used as the starting materials for the synthesis of nanostructured highly photoactive TiO₂-based materials by a simple thermal annealing process [19–21]. TiO₂ Degussa P25 with an anatase/rutile ratio of 80/20 is a widely used commercial photocatalyst because of its relatively high level of activity in many photocatalytic reaction systems [22]. Actually, it is not easy to find a photocatalyst showing activity higher than that of TiO₂ P25, so it has been used as a standard TiO₂ photocatalyst [23–25]. Developing a material as a reasonable alternative for TiO₂ P25 is the basic goal for the development of novel photocatalyst.

TiO₂ contacted with large work function metal nanoparticles results in Schottky barrier at the interface. At first, a consequence of photopromoted electrons preferentially migrates from the conduction band of TiO₂ to the metal nanoparticles, thus decreasing electron–hole recombination. Then, the efficient charge separation was induced by the Schottky interface between metal nanoparticles and n-type semiconducting TiO₂. Finally, protons formed by the reaction between holes and water are reduced by electrons injected previously into the metal nanoparticles. This is due to electrons being restricted in the metal nanoparticle by the Schottky barrier, preventing the undesired electron–hole recombination [26–29]. It has been found to enhance the photocatalytic performance by decorations of TiO₂ with carbon nanotubes [30], graphene [31], metals, and metal oxides such as Pt [27,32,33], Au [34], Pd [27,32,34], PdO [35], Ni [36], and Ag [37,38]. However, using noble metal to decorate TiO₂ catalysts is still an effective method to obtain the high activity photocatalysts. Ismail et al. reported the synthesis of a highly active photocatalyst of mesoporous PdO–TiO₂ material using different PdO concentration through simple one-step sol–gel reactions. Pd²⁺-ions have been immobilized into TiO₂ networks by cross linking triblock copolymer as the structure-directing agents to develop highly efficient PdO–TiO₂ photocatalysts. The measured photonic efficiency is as high as 19.5% [35]. Besides, Roy et al. reported a synthesis of Ti_{1-x}Pd_xO_{2-δ} crystallizing in anatase phase with Pd²⁺-ion, showing that enhanced CO oxidation at Pd²⁺ ion site and O₂ or NO photo dissociation at oxide ion vacancy is responsible for the enhanced catalytic activity [39]. Palladium (Pd) is the cheapest and most commonly used metal among the noble metals for co-catalyst with the TiO₂ photocatalyst, and the cost of Pd metal is approximately 20–25% of that of platinum (Pt) metal.

Thermal treatments carried out in hydrogen for co-catalyst could be an useful method for enhancing the photocatalytic activity, but it also could influence the crystalline and electronic structure of TiO₂ thus inducing a bandgap narrowing [40,41]. Hence, the reduce process could cause the change

between TiO₂ and metal nanoparticles at the same time. However, highly-crystalline TiO₂ usually could not be reduced by hydrogen in normal pressure [42].

Herein, in order to prepare the catalytically active TiO₂ NFs from SHT NFs, a quick screening was carried out to find the appropriate calcination temperature. The obtained anatase-rutile TiO₂ NFs calcined at 800 °C shows the highest photocatalytic activity among the thermally treated TiO₂ NFs. Then, we prepared palladium decorated TiO₂ NFs (TiO₂-Pd NFs) by wet impregnation method followed by hydrogen-thermal reduction process in order to enhance the photocatalytic activity. The obtained TiO₂-Pd NFs reduced at 300 °C might be a reasonable alternative for the traditional TiO₂ P25 for photocatalytic applications in the decomposition of organic dyes and photocatalytic hydrogen generation.

Experimental details

Preparation of palladium decorated TiO₂ nanofibers

For the preparation of sodium hydrogen titanate nanofibers (SHT NFs), we suspend 2.50 g TiO₂ anatase powder (Aldrich, 98%) in 62.5 mL of 10.0 M NaOH aqueous solution, followed by a treatment in a Teflon-lined autoclave at 150 °C for 24 h, applying revolving around its short axis. Then, the product, sodium titanate nanofibers, was then washed in 0.10 M HCl to exchange sodium ions for protons. The neutralized product was washed with deionized water and finally filtered and dried in the air at 70 °C. In order to prepare the TiO₂ NFs, the SHT NFs were calcined in the air at various temperatures, 400, 550, 600, 700, 800 and 950 °C, respectively, with heating rate of 5 °C/min for 4 h to find the optimal calcination temperature. For TiO₂-Pd NFs, 29.2 mg of palladium(II) acetylacetonate (Aldrich, 99%) was dissolved in 200 mL of acetone and mixed with 1.0 g of TiO₂ NFs by ultrasonic agitation for 3 h. Then, the mixture was stirred for 6 h. After evaporating the solvent at ~80 °C under N₂ atmosphere, the samples were calcined in the air at 300 °C for 2 h, and then reduced at various temperature, such as 300, 400 and 500 °C, in 15% H₂ (in N₂ buffer) flow for 4 h to obtain TiO₂-Pd NFs with ~1.0 wt% metal loading.

Characterization

In order to obtain the Raman scattering spectra, anatase TiO₂ (Acros, TiO₂, anatase powder, 98+%), rutile TiO₂ (Alfa Aesar, TiO₂, rutile powder, 99.5%) and various thermally treated TiO₂ NFs were positioned on a high-resolution piezoelectric stage of the scanning microscopy (WITec, Alpha300S, Germany) and excited by a 632.8 nm He–Ne laser. The laser beam was focused with a 10X objective lens (Nikon plane objective, NA ~ 0.9), and the diameter of the laser beam focus was about 10 μm. The crystal structures of various thermally treated TiO₂ NFs were determined by X-ray diffraction spectrometry (XRD) (Rigaku, TTRAXIII, Japan) using Cu K_α radiation at 50 kV and 300 mA. XRD patterns were collected from 2θ between 10 and 80 with a 0.005° step at 5 min⁻¹. The microstructure and morphology of various thermally treated TiO₂ NFs were studied by scanning electron microscopy (SEC, SNE-4500M MiniSEM, Korea).

Photocatalytic activity testing

The photodegradation of organic dye behaviors of TiO₂ P25 (Degussa) and TiO₂-based materials were performed by monitoring the decoloration of brilliant green. In a typical experiment, 20.0 mg of catalyst was sonicated for 5 min in 150 mL brilliant green aqueous solution (20 mg/L). The suspension was irradiated with UV-B light (Sankyo Denki, G8T5E, 8 W × 6 sets) under vigorous stirring at ambient conditions. After a centrifuging process (15 min at 3500 rpm), the UV–Vis spectrum of the remained brilliant green and its derivatives in the supernatant was recorded in the spectral range from 400 nm to 900 nm (Jasco Analytical Instruments, V-650 UV–Vis Spectrophotometer). The brilliant green concentration was calculated from the absorbance at $\lambda = 624$ nm, extrapolated to a previously plotted calibration curve.

Hydrogen generation from ethanol–water mixtures

Photocatalytic water splitting hydrogen generation tests were carried out by using 1:1 volume ratio mixture of ethanol and water (total 2.0 L), in which 10.0 mg of TiO₂-based catalyst was suspended before each experiment. The temperature of the mixture was kept near the room temperature. For light source, six pieces of UV-B lamps (Sankyo Denki G15T8E UV-B lamp, the wavelength of maximum emission of UV-B lamp was ~312 nm, and the power was 8.0 W) were placed in a hexagonal arrangement around the reactor. To avoid sedimentation of the catalyst powders, nitrogen (99.995%) was bubbled through the reactor with a flow rate of 400 mL/min, serving also as a purging gas for the evolving gaseous products. The outlet of the reactor was connected a cold trap, a molecular sieve, and a hydrogen analyzer (Status Scientific Controls Ltd, FGD3).

Results and discussion

In order to prepare the high catalytically TiO₂ NFs from SHT NFs, a quick screening was carried out to find the suitable calcination temperature. Anatase TiO₂, rutile TiO₂ and SHT NFs calcined at various temperatures were identified by Raman spectrometer. The Raman spectra of anatase TiO₂ and rutile TiO₂ are as standard. The spectra presented several TiO₂ phases, such as anatase, rutile, and mixed phase, as shown in Fig. 1. The TiO₂ NFs samples prepared from SHT NFs calcined at various temperatures for 4 h were denoted as TiO₂-xxx NFs, for which “xxx” indicating various calcination temperatures. Raman spectra showed that samples calcined below 400 °C presented a mixed phase of amorphous phase and anatase. When the calcination temperature went above 550 °C, the crystal phase transformed to anatase. The anatase TiO₂ showed major Raman bands at 144, 200, 398, 515, 517 and 640 cm⁻¹, but the Raman band at 515 cm⁻¹ was superimposed with the 517 cm⁻¹ band. These bands can be attributed to the six Raman-active modes of anatase phase with the symmetries of E_g, E_g, B_{1g}, A_{1g}, B_{1g}, and E_g, respectively [43]. When calcination temperature went above 800 °C, TiO₂ NFs sample presented rutile phase with their major Raman bands at 448 and 617 cm⁻¹. For rutile TiO₂, the typical Raman bands appear

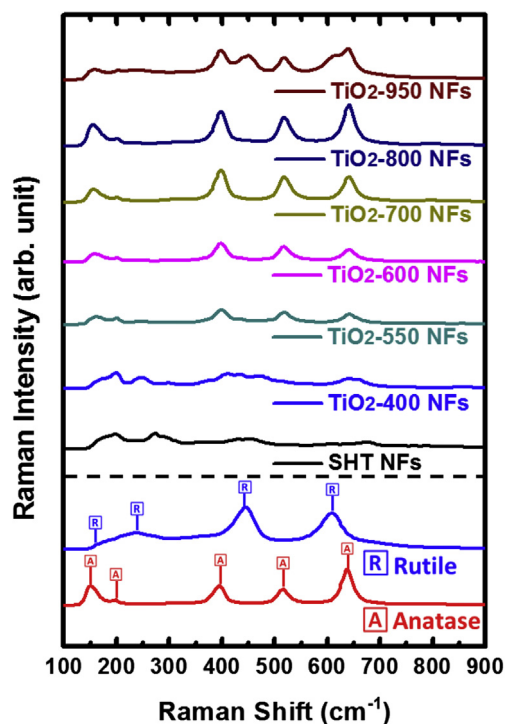


Fig. 1 – Raman spectra of anatase TiO₂, rutile TiO₂, SHT NFs and SHT NFs calcined at various temperatures, 400, 550, 600, 700, 800 and 950 °C, for 4 h.

at 143, 243, 448 and 617 cm⁻¹, which can be ascribed to the B_{1g}, two-phonon scattering, E_g, and A_{1g} modes of rutile phase, respectively [43].

The XRD pattern of the synthesized SHT NFs is similar to our previous work [27,32]. SHT NFs treated with 4 h of various calcination temperatures in the air were measured by XRD as shown in Fig. 2(a). The X-ray diffraction patterns of the thermally treated SHT NFs showed an obvious phase transformation. With calcination temperatures of up to 550 °C, there was only formation of anatase phase. The intensity of reflections increased as the calcination temperature increased, indicating better ordering of the lattice. However, at calcination temperatures of above 800 °C, the formation of TiO₂ rutile phase started, as shown by the appearance of its (110) reflection at ~27.4° of 2 θ (Fig. 2(b)). The crystalline structure of TiO₂-800 NFs and TiO₂-950 NFs are the mixture of anatase and rutile phase. The XRD result was consistent with the results of Raman measurement.

The photocatalytic activities of the synthesized TiO₂ NFs were tested with UV-B light-induced photodegradation of brilliant green in aqueous solution. UV–vis spectra of brilliant green as a function of UV light irradiation time were recorded. From the absorbance measured at $\lambda = 624$ nm, the corresponding dye concentrations were calculated using a calibration curve measured previously. The overall order of photodegradation activity was TiO₂-800 NFs > Degussa P25 > TiO₂-700 NFs > TiO₂-600 NFs > TiO₂-950 NFs > TiO₂-400 NFs as shown in Fig. 3(a), indicating that TiO₂-800 NFs had the highest photodegradation activity. TiO₂-800 NFs containing

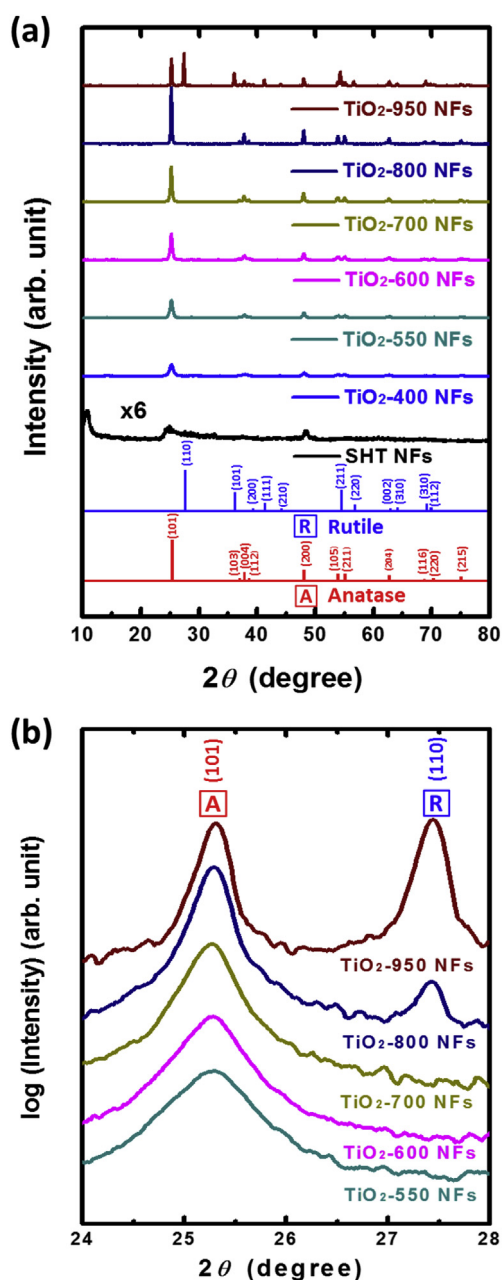


Fig. 2 – X-ray diffraction patterns of SHT NFs and SHT NFs calcined in the air at 400, 550, 600, 700, 800 and 950 °C for 4 h (A = Anatase; R = Rutile).

anatase-rutile mixed phase exhibited the highest photodegradation activity due to the efficient charge separation at the interface between anatase TiO_2 and rutile TiO_2 . Because the energy levels of both conduction band and valence band on rutile TiO_2 are higher than those on anatase TiO_2 , it generated a barrier between anatase TiO_2 and rutile TiO_2 , which can separate photogenerated electrons and holes easily. On the other hand, the electron–hole recombination rates of single phase TiO_2 are larger than TiO_2 with anatase-rutile mixed phase. TiO_2 -catalyzed photodegradation of different dyes essentially follow Langmuir–Hinshelwood

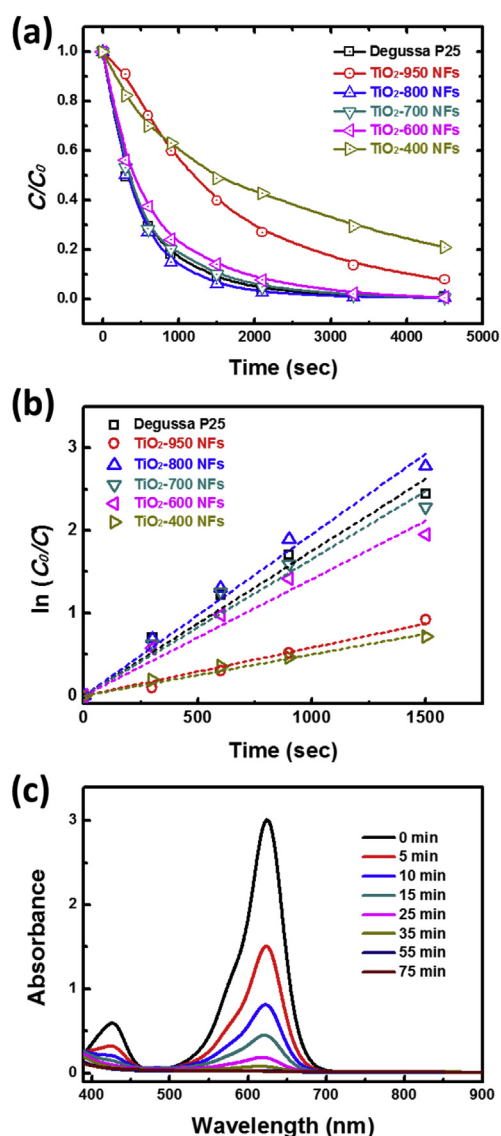


Fig. 3 – (a) Photocatalytic activities of TiO_2 NFs calcined at various temperatures over the photodegradation of brilliant green under UV-B irradiation. (b) Linearized kinetic plots for the degradation of brilliant green using various types of TiO_2 NFs based catalysts. (c) Absorbance spectra of brilliant green (initial concentration of 20.0 mg/L) as a function of illumination time with TiO_2 -800 NFs (20.0 mg dispersed in 150 mL of solution). (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

kinetics, which can be simplified to an apparent first-order kinetics at lower initial brilliant green concentrations, mathematically described as $\ln(C_0/C) = kt$; where C is the concentration of the dye at time t , C_0 is the initial concentration, and k is the apparent reaction rate constant [27,44]. Plotting the logarithm of the reciprocal of brilliant green concentrations as a function of time, we obtained linear slopes for each catalyst studied and they were in good agreement with the

Langmuir–Hinshelwood model. TiO₂-800 NFs and Degussa P25 gave the fast decoloration phenomenon for photodegradation of brilliant green, and their calculated rate constants were $\sim 1.96 \times 10^{-3} \text{ s}^{-1}$ and $\sim 1.75 \times 10^{-3} \text{ s}^{-1}$, respectively. In Fig. 3(b), TiO₂-800 NFs showed higher photocatalytic activity than Degussa P25. TiO₂-800 NFs consisting a mixed phase of anatase and rutile may be a reasonable alternative of the

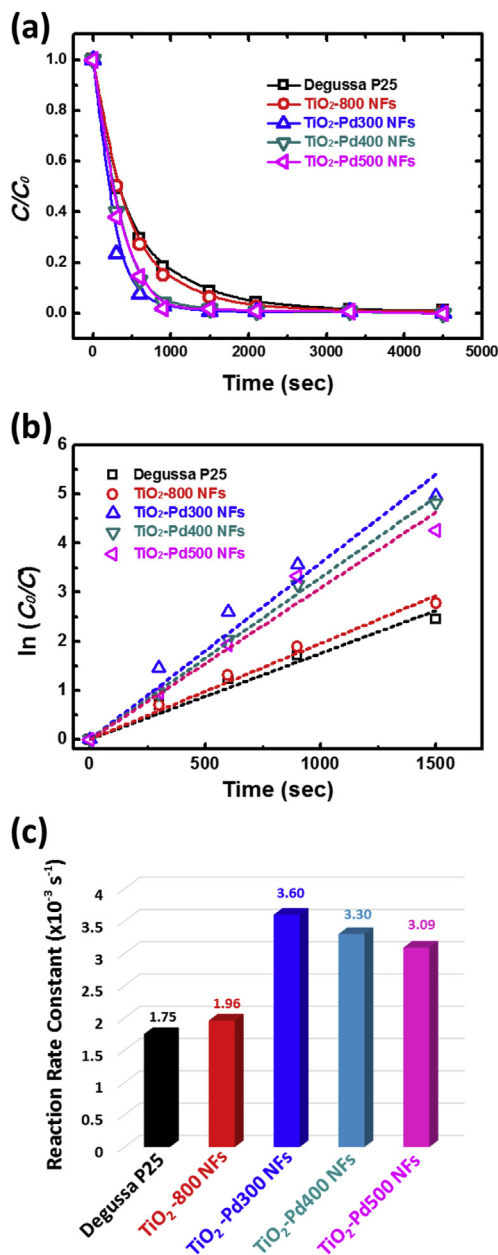
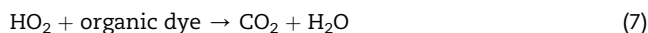
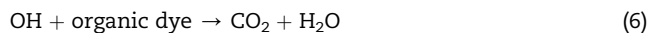


Fig. 4 – (a) Photocatalytic activities of TiO₂-Pd NFs reduced at various temperatures over the photodegradation of brilliant green under UV-B irradiation. (b) Linearized kinetic plots for the degradation of brilliant green using various types of TiO₂-Pd NFs based catalyst. (c) The bar charts of photodegradation reaction rate constants of P25, TiO₂ NFs and various TiO₂-Pd NFs based catalysts. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

traditional Degussa P25 for photocatalytic applications. The UV–vis absorbance spectra for UV-B light-induced photodegradation of brilliant green in aqueous solution as a function of illumination time with TiO₂-800 NFs were shown in Fig. 3(c).

In order to enhance the performance of photodegradation toward organic dye and photocatalytic hydrogen generation, TiO₂-800 NFs decorated with Pd-based nanoparticles was prepared because of palladium is the cheap and most commonly used metal among the noble metals for co-catalyst with TiO₂ photocatalyst. The mechanism of photodegradation of brilliant green in aqueous solution is shown in Eqns. (1)–(7) [2,45].



Samples prepared from Pd-based nanoparticles decorated TiO₂-800 NFs were reduced at various temperatures for 4 h. These samples were denoted as TiO₂-Pdyyy NFs, for which “yyy” indicating various reduction temperatures of 300, 400, and 500 °C, respectively. Photocatalytic activities of TiO₂-Pd NFs reduced in N₂/H₂ atmosphere at various temperatures were measured. The results were shown in Fig. 4(a). Among Pd

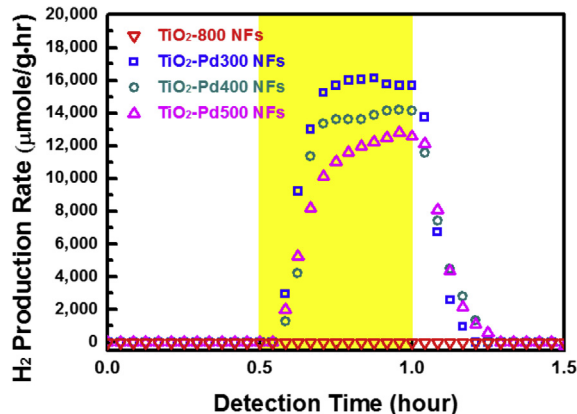


Fig. 5 – Photocatalytic hydrogen production rates of TiO₂-800 NFs and TiO₂-Pd NFs with various reduction temperatures under the irradiation of UV-B lamp.

decorated TiO₂ NFs based catalysts, TiO₂-Pd300 NFs exhibited the highest photodegradation activity of brilliant green with a calculated rate constant of $\sim 3.60 \times 10^{-3} \text{ s}^{-1}$ as shown in Fig. 4(b). The photocatalytic activity decreased as the reduction temperature increased for the TiO₂-Pd NFs based catalysts. Hence, the reduction temperature of 300 °C is enough for preparing the TiO₂-Pd NFs based catalyst with high

photocatalytic performance, because of the chemical state of palladium plays an important role for the preparation of high performance catalysts. TiO₂-Pd300 NFs is suitable for the applications of photodegradation of brilliant green, and its UV-vis absorbance spectra of brilliant green in aqueous solution for UV-B light-induced photodegradation as a function of illumination time were shown in Fig. 4(c).

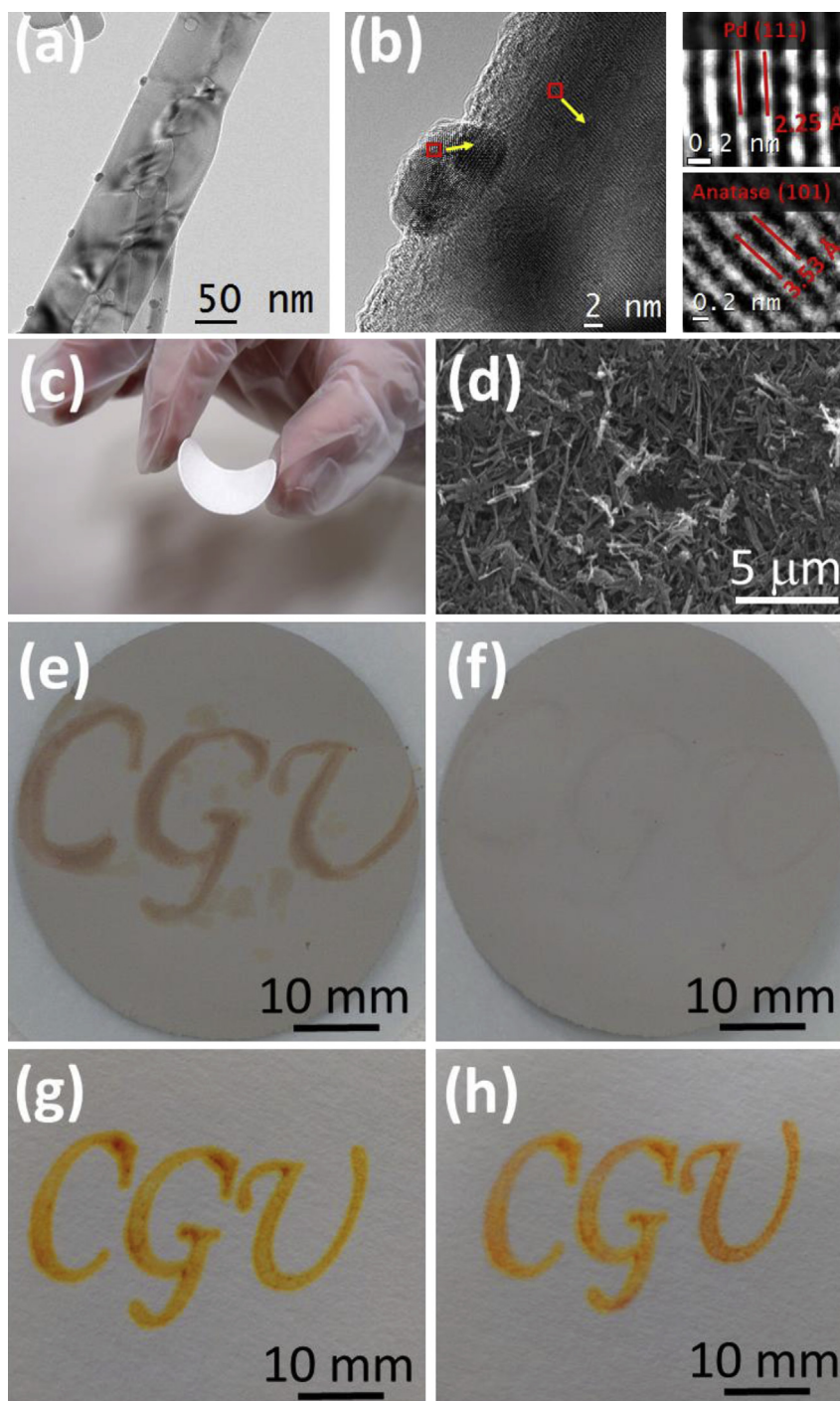


Fig. 6 – (a) TEM images of TiO₂-Pd300 NFs. (b) The high-resolution TEM image of TiO₂-Pd300 NFs, and the high-magnification lattice images of Pd-based nanoparticles (top) and TiO₂ NFs (bottom). (c) Photograph of a bent TiO₂-800 NFs composite membrane, and (d) SEM image of cellulose/TiO₂-Pd300 NFs surface. The photographs of three letters “CGU” affixed on (e) the composite membrane, (f) the composite membrane after 1.0 h of UV-B exposure, (g) a piece of paper, and (h) a piece of paper after 1.0 h of UV-B exposure.

Photocatalytic hydrogen production rates of TiO₂-800 NFs and TiO₂-Pd NFs with various reduction temperatures, such as 300, 400 and 500 °C, under the irradiation of UV-B lamp are shown in Fig. 5. For pure TiO₂ materials without metal decoration, it is not active in the photocatalytic hydrogen generation, and must be combined with a precious transition metal to yield an active photocatalyst. The purpose of metal nanoparticle decorated on TiO₂ surface is to generate a Schottky barrier at the interface between TiO₂ and metal nanoparticles. The highest hydrogen production rate was found with TiO₂-Pd300 NFs under UV-B irradiation with a rate of as high as 16,200 μmol/g·hr. Moreover, the photocatalytic hydrogen production rate was found to be around 14,300 μmol/g·hr for TiO₂-Pd400 NFs, and 12,900 μmol/g·hr for TiO₂-Pd500 NFs. The photocatalytic hydrogen production rate decreased as the reduction temperature increased for TiO₂-Pd NFs based catalysts. In our previous study [46], the chemical state of Pd (Pd, PdO, and Pd²⁺-ion) plays an important role for photocatalytic hydrogen production, because of its ability to capture the photogenerated electrons effectively and to reduce the rate of electron-hole recombination. The hydrogen reduction time increase, the amount of Pd and PdO increase, but the amount of Pd²⁺-ion decrease. Pd-based nanoparticle decorations with the appropriate chemical states on TiO₂-Pd300 NFs surface could capture the photogenerated electrons effectively and could reduce the rate of electron-hole recombination to enhance the photocatalytic hydrogen production rate. This is due to electrons being restricted in the metal nanoparticle by the Schottky barrier, preventing the undesired electron-hole recombination.

TEM images of TiO₂-Pd300 NFs were shown in Fig. 6(a). The length of TiO₂ NFs was up to a few micrometers and its diameter was ranged from 80 to 150 nm. Pd-based nanoparticles distributed on the surface of TiO₂ NFs presented uniform size distribution with average particle diameters of 7.3 ± 1.3 nm, and (111) crystal plane is 2.25 Å as shown in Fig. 6(b). The (101) crystal plane of TiO₂ was observed for TiO₂-Pd300 NFs, and the (101) spacing is 3.53 Å. Next, we used cellulose and the TiO₂-800 NFs to prepare composite films that enabled handling for easy-recycling applications as shown in Fig. 6(c). The presence of TiO₂ with different shapes in the different types of catalyst materials can influence mechanical friction and sticking of adjacent cellulose fibers coated with the nanoparticles, thus, affecting flexibility. Fig. 6(d) showed the cellulose microfibers were uniformly coated with the TiO₂ NFs. Using photocatalysts to degrade organic compounds on surfaces has attracted considerable attention in recently years, due to the possible applications in self-cleaning coatings and antimicrobial surfaces. Here, we also used cellulose and TiO₂-Pd300 NFs to prepare freestanding composite membranes that enabled easy handling. Using the photocatalyst composite films we demonstrated degradation of organic dyes deposited on the surface of thin catalyst membranes as shown in Fig. 6(e-h). Three letters “CGU” were affixed on a TiO₂-Pd300 NFs composite film (Fig. 6(e)) and a piece of paper (Double A, 80 gsm) (Fig. 6(g)) by a stamp with a highlight pen ink (Simbalion, FM-35), respectively. Then, these two samples were exposed to UV-B light for 1.0 h. The composite membranes consisting of cellulose and TiO₂-Pd300 NFs showed

obvious changes in the intensity of the color pattern as shown in Fig. 6(f). The original stain could hardly be seen, indicating that the fading of the dye was practically complete. However, it just exhibits minor changes for the “CGU” on the paper for the stain on the paper, as shown in Fig. 6(h), because no catalyst to improve the photocatalytic degradation of organic dye.

Conclusion

In summary, this study included: (i) photodegradation of brilliant green with TiO₂ NFs and TiO₂-Pd NFs, (ii) photocatalytic hydrogenation of TiO₂ NFs and TiO₂-Pd NFs, and (iii) degradation of organic dyes with freestanding membranes containing cellulose and TiO₂ NFs. At first, SHT NFs calcined at 800 °C for 4 h leads to the formation of anatase-rutile TiO₂ NFs. Compared with other TiO₂ based catalysts, the brilliant green decoloration using TiO₂-Pd300 NFs shows the highest photocatalytic activity. In addition, TiO₂-Pd300 NFs also exhibits the highest H₂ production rate at 16,200 μmol/g·hr. In both cases, TiO₂-Pd300 NFs has proven to be more efficient. This can be explained by the formation of nano-sized Schottky interfaces at the contacts between TiO₂ and metal nanoparticles. Furthermore, the ability to fabricate cellulose/TiO₂-Pd300 NFs composite membrane was developed for utilizing photocatalyst materials in freestanding films. The obtained TiO₂-Pd300 NFs may be a reasonable alternative of the commercial Degussa P25 for photocatalytic applications in the decomposition of organic dyes and photocatalytic hydrogen generation.

Acknowledgments

Financial support obtained from Ministry of Science and Technology of Taiwan (MOST 102-2633-E-182-001, MOST 103-2221-E-182-008-MY2 and MOST 104-3113-E-002-010) and Chang Gung University Research Project is highly appreciated. We appreciate Prof. Wei-Fang Su at National Taiwan University and Dr. Ming-Tao Lee group (BL-13A1) at National Synchrotron Radiation Research Center for useful discussion and suggestions.

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